



Numerical simulation of metal vapor behavior in arc plasma



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ABSTRACT

A gas tungsten arc in helium is modeled taking into account the contamination of the plasma by the metal vapor from the weld pool surface. The whole region of gas tungsten arc welding, namely the tungsten cathode, arc plasma and weld pool, is treated using a unified numerical model. A viscosity approximation is used to express the diffusion coefficient in terms of the viscosities of the helium gas and the iron vapor. The time-dependent two-dimensional distributions of temperature, velocity and iron vapor concentration are predicted, together with the weld penetration as function of time for a 150 A arc current at atmospheric pressure. It is shown that the thermal plasma in gas tungsten arcs is markedly influenced by iron vapor from the weld pool surface and that the concentration of the iron vapor in the plasma is dependent on temperature of the weld pool surface.

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1. Introduction

During the arc welding process, four states of matter, solid, liquid, gas and plasma, simultaneously exist and mutually interact within a volume of only 1 cm³. The temperature range is wide, ranging from about 20 000 K in the arc plasma, about 3000 K in the tungsten cathode, about 2000 K in the molten steel, down to room temperature in the surrounding regions [1]. Due to the remarkable progress in computer simulation and observation techniques recently, it has become possible to understand the phenomena in arc welding processes quantitatively [2–4]. However, it has not been possible to accurately predict the welding parameters, such as the arc voltage and the weld geometry. It is known empirically that the arc voltage in gas tungsten arc (GTA) on water-cooled copper differs from that in GTA welding. This phenomenon is caused by the metal vapor from the weld pool surface. For a full understanding and accurate prediction of these parameters, it is necessary to understand the behavior of metal vapor in the arc plasma.

Metal atoms generally have more low-energy excited states, and are more easily ionized, than atoms of shielding gases such as argon and helium. These characteristics contribute to an increase of the radiative emission coefficient and the electric conductivity of the plasma. It is estimated that the former affects the thermal pinch effect and the energy efficiency of the arc plasma and that the latter affects the current density distribution. Tashiro et al. [5] conducted a virtual experiment by numerically simulating a pure helium arc and an arc in

helium uniformly mixed with 30 mol% iron atoms, and showed that an obvious arc constriction occurred for the latter case. Furthermore, they reported that the energy efficiency greatly decreased from about 80% to about 35%. These results suggested that existence of metal vapor changed the heat source property in the arc welding process, and consequently changed the size and the shape of the molten pool.

As noted above, there have been significant advances in the simulation of the arc welding. For example, a numerical simulation of gas metal arc welding, including the formation of droplets from a steel wire electrode, has been reported [2]. However, the welding gas was pure argon, and the effect of metal vapor was not considered. On the other hand, calculations of the behavior of metal vapor in an atmospheric pressure plasma have also been reported [6]. However, the conditions were far from those of welding because a solid electrode with constant temperature was assumed. It is important for accurate understanding of the arc welding process to consider the mixing of the metal vapor in a model that takes into account the tungsten cathode, the arc plasma and the weld pool.

In the present paper, we use a numerical model of stationary GTA welding taking into account the iron vapor produced from the weld pool surface, and we simulate the distribution of the metal vapor, the plasma temperature and the formation of the weld pool in GTA welding.

2. Simulation model

The tungsten cathode, arc plasma and anode are described relative to cylindrical coordinates, assuming rotational symmetry around the arc axis. The calculation domain is shown in Fig. 1. The diameter of the tungsten cathode is 3.2 mm with a 60° conical tip. The anode is of

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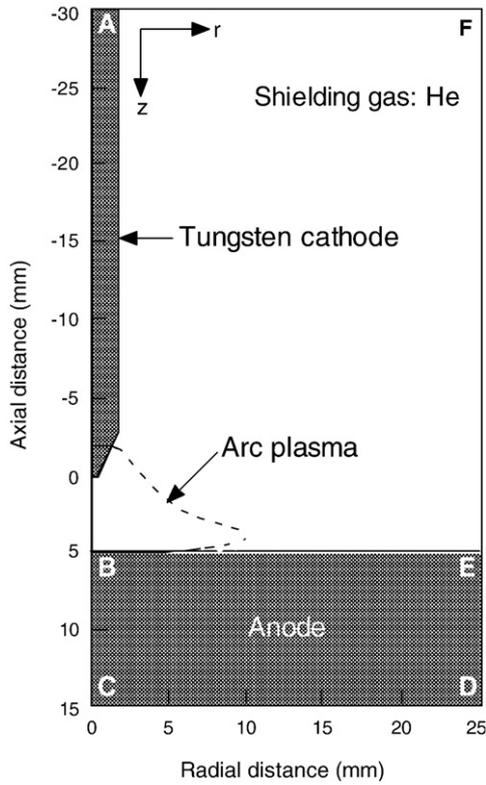


Fig. 1. Schematic illustration of simulation domain.

water-cooled copper or stainless steel. Helium shielding gas is introduced from the outside of the cathode on the upper boundary at the flow rate of 30 L/min.

The convection in the weld pool is influenced by the shear stress due to the convective flow of the cathode jet, the Marangoni force due to the gradient in the surface tension of the weld pool, buoyancy due to gravity and the electromagnetic pinch force due to the arc current. Only the driving forces of the weld pool convection at the boundary between the weld pool and the arc plasma are explained here. First, the shear stress is already included in radial momentum conservation

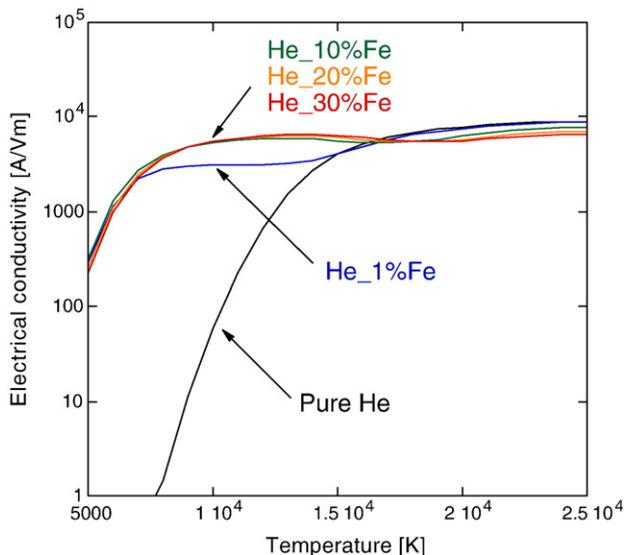


Fig. 2. Dependence of electrical conductivities of helium gas on temperature for each mixing ratio.

through the viscosity at the anode surface. Second, the Marangoni force is given by [7]

$$M_A = \frac{\partial}{\partial z} \left(\frac{\partial \gamma}{\partial T} \frac{\partial T}{\partial r} \right) \quad (1)$$

where T is temperature, γ is the surface tension of the weld pool. In this paper, stainless steel is assumed to have low sulfur content (about 10 ppm) and the variation of the surface tension at the weld pool surface is assumed to decrease linearly with increasing temperature ($\partial \gamma / \partial T = -0.46$ mN/mK) [7].

A species conservation equation expressed by Eq. (2) is applied to take into account the metal vapor behavior [6]. In practice, metal vapor species in an arc plasma with a stainless steel anode include Fe, Cr, Ni, Mn and so on [8]. However, to simplify the model and facilitate calculation, only iron vapor is considered in this model.

$$\frac{\partial}{\partial t} (\rho C_1) + \frac{1}{r} \frac{\partial}{\partial r} (r \rho v_r C_1) + \frac{\partial}{\partial z} (\rho v_z C_1) = \frac{1}{r} \frac{\partial}{\partial r} \left(r \rho D \frac{\partial C_1}{\partial r} \right) + \frac{\partial}{\partial z} \left(\rho D \frac{\partial C_1}{\partial z} \right) \quad (2)$$

where t is time, v_r and v_z are the radial and axial velocities, ρ is the density, C_1 is mass fraction concentration of iron vapor and D is the binary diffusion coefficient, which is expressed by the viscosity approximation equation:

$$D = \frac{2\sqrt{2}(1/M_1 + 1/M_2)^{0.5}}{\left\{ (\rho_1^2/\beta_1^2\eta_1^2M_1)^{0.25} + (\rho_2^2/\beta_2^2\eta_2^2M_2)^{0.25} \right\}^2} \quad (3)$$

where M_1 and M_2 are the molecular weights of iron and the shielding gas respectively. Similarly, ρ_1 , ρ_2 , η_1 , η_2 are respectively the density and viscosity of iron and the shielding gas. β_1 , β_2 are the dimensionless constant defined as $\beta_i = (D_{if}\rho_i)/\eta_i$, and theoretically ranged from 1.2 to 1.543 for various species of gas, such as Ar, He, H₂, N₂, O₂, CO₂ and so on. It is assumed that $\beta_1 = \beta_2 = 1.385$, which is based on the mean value of experimental data [9]. The viscosity approximation is not strictly justified since it does not take into account ionized species and is at best reasonably accurate [10]; however it is considered to be a useful first approximation for the arc welding model.

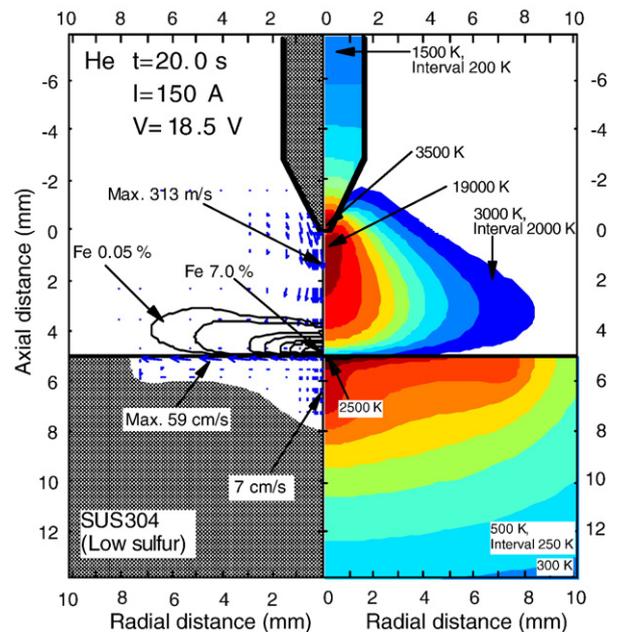


Fig. 3. Calculated results for helium GTA welding for 150 A at 20 s after arc ignition.

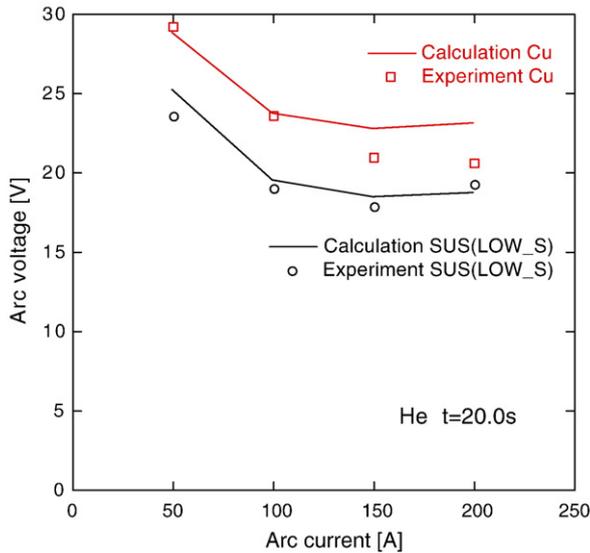


Fig. 4. Comparison of calculated arc voltage with experimental results.

C_1 is set to be zero in the cathode area and in the solid area of the anode. However, at the anode surface where the temperature is above the melting point, C_1 is set to [6]:

$$C_1 = \frac{p_{v,1}M_1}{p_{v,1}M_1 + (p_{\text{atm}} - p_{v,1})M_2} \quad (4)$$

where p_{atm} is atmospheric pressure and $p_{v,1}$ is the iron vapor partial pressure, which is a function of the weld pool temperature [11]. According to Eq. (4), C_1 has values between zero and 1.0. For other boundary conditions, $C_1 = 0$ at AF and FE shown in Fig. 1, and $(\partial C_1 / \partial r) = 0$ at the arc axis (AB).

In the present model, plasma properties are dependent on not only the temperature but also the mole fraction of iron vapor. Plasma properties at intermediate concentrations of iron vapor are calculated using a linear approximation based on the properties at 0 mol%, 1 mol%, 10 mol%, 20 mol% and 30 mol% [12]. The properties were calculated assuming the arc plasma to be in the local thermodynamic equilibrium (LTE), and using the Chapman–Enskog approximation [12]. For example, the electrical conductivities, which are significantly affected, are shown in Fig. 2. The electrical conductivities are greatly increased by the addition of iron vapor at temperatures below 15000 K, and the values for mixing ratios 1%, 10%, 20% and 30% are almost the same.

The other approximations, governing equations and boundary conditions are given in detail in our previous papers [5,13]. The governing and auxiliary equations are solved iteratively by the SIMPEC numerical procedure.

3. Calculation results

The present model is applied to the case of stationary helium GTA welding of stainless steel. Fig. 3 shows the two-dimensional distribution of temperature, fluid flow velocity and mole fraction of iron vapor at a time 20 s after arc ignition, namely, when the weld pool has grown to a reasonable extent. The distribution of iron vapor depends on the diffusion term and the convection term, as described in Eq. (2). Due to the cathode jet, which leads to flow velocities of over 300 m/s in the welding arc, the convection term has a strong effect. Therefore, it is found that distribution of iron vapor expands in the radial direction and is concentrated around the weld pool surface. The concentration of iron vapor in the arc plasma is up to 7 mol%, which is in good agreement with the experimental data presented by Terasaki [8].

Fig. 4 shows calculated values of arc voltage for 50 A, 100 A, 150 A and 200 A arc current. Experimental results are also shown (the calculated

results are represented by lines and the experimental results by symbols). Arc voltages for a water-cooled copper anode and a stainless steel anode are shown; in the case of stainless steel, arc voltages are again shown at 20 s after arc ignition. The experiment is consistent with the calculation. The work piece is the SUS304 stainless plate (50 mm × 100 mm × 10 mm thick) with 10 ppm sulfur content mounted on a water-cooled copper plate. The welding conditions, such as arc length, electrode diameter and flow rate of shielding gas, are the same as the conditions of the calculation. Fig. 4 shows that the arc voltages for the stainless steel anode are about 5 V lower than those for the water-cooled copper anode at any arc current. This is due to the effect of iron vapor from the weld pool. As shown in Fig. 2, even a small amount of iron vapor increases the electrical conductivity. Consequently, the arc voltage decreases because of the lower voltage gradient, particularly just above the weld pool. Between 100 and 200 A, a difference of 2 to 3 V is found between calculation and experiment in the case of water-cooled copper anode. A likely reason for this difference is that a small amount of copper vapor is produced from the anode surface at high current, for which the heat input density is large, thus decreasing the voltage in the experiments.

Fig. 5 shows current density distributions at the anode surface. For the stainless steel anode, the distribution calculated neglecting the effect of iron vapor on the arc plasma is also given, to clearly show the influence of iron vapor. The peak current density is the largest, over 5000 A/cm², for the case of the water-cooled copper anode. The distribution for the stainless steel anode, calculated taking into account the iron vapor is clearly different from that neglecting the iron vapor. The peak current density in the latter case is over 3500 A/cm², while the presence of iron vapor decreases the peak current density to 1100 A/cm², and the region through which current flows is expanded by the increase of the electric conductivity at lower temperatures; as shown in Fig. 2, the increase is greater at the lower temperatures that occur away from the arc axis.

4. Conclusions

We have used a numerical model of a stationary GTA welding taking into account the iron vapor from the weld pool surface, and have simulated the effect of the iron vapor on the properties of the arc and weld pool. The main conclusions are summarized as follows:

- (1) Due to the cathode jet velocity of over 300 m/s in the welding arc, the convection term strongly affects the distribution of the

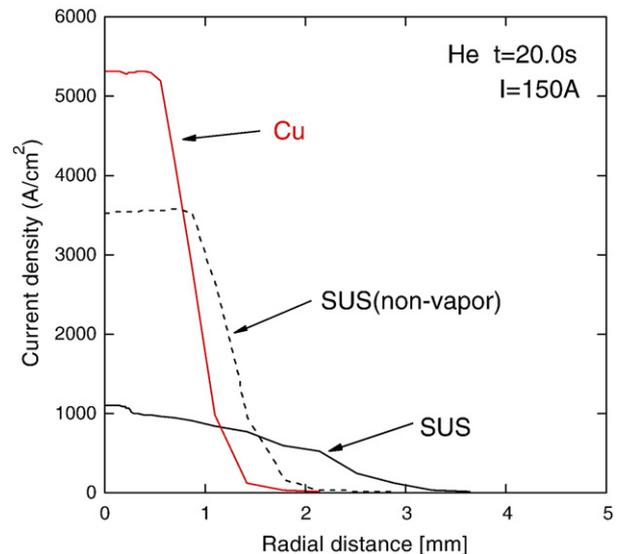


Fig. 5. Comparison of calculated current densities at the anode surface.

iron vapor. It was found that iron vapor expanded mainly in the radial direction and remained concentrated around the weld pool surface.

- (2) Calculated values of arc voltage were in good agreement with experimental results, and the arc voltage for a stainless steel anode was clearly lower than that for a water-cooled copper anode.
- (3) The arc current distribution for a stainless steel anode calculated taking into account the iron vapor was clearly different from that for calculated neglecting the iron vapor. The peak current density obtained neglecting the iron vapor was over 3500 A/cm², while that obtained taking into account the iron vapor decreased to about 1100 A/cm². In the latter case, the region through which current flows is expanded by the increase of the electric conductivity at lower temperatures, which is particularly important away from the arc axis.

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